Investigation of Instability in High Jc Nb3Sn Strands

L. Cooley

January 2005

Superconducting Magnet Division
Brookhaven National Laboratory

U.S. Department of Energy
USDOE Office of Science (SC), High Energy Physics (HEP) (SC-25)

Notice: This technical note has been authored by employees of Brookhaven Science Associates, LLC under Contract No.DE-AC02-98CH10886 with the U.S. Department of Energy. The publisher by accepting the technical note for publication acknowledges that the United States Government retains a non-exclusive, paid-up, irrevocable, world-wide license to publish or reproduce the published form of this technical note, or allow others to do so, for United States Government purposes.
DISCLAIMER

This report was prepared as an account of work sponsored by an agency of the United States Government. Neither the United States Government nor any agency thereof, nor any of their employees, nor any of their contractors, subcontractors, or their employees, makes any warranty, express or implied, or assumes any legal liability or responsibility for the accuracy, completeness, or any third party’s use or the results of such use of any information, apparatus, product, or process disclosed, or represents that its use would not infringe privately owned rights. Reference herein to any specific commercial product, process, or service by trade name, trademark, manufacturer, or otherwise, does not necessarily constitute or imply its endorsement, recommendation, or favoring by the United States Government or any agency thereof or its contractors or subcontractors. The views and opinions of authors expressed herein do not necessarily state or reflect those of the United States Government or any agency thereof.
Investigation of instability in high \( J_c \) Nb\(_3\)Sn strands

A. K. Ghosh, L. D. Cooley, and A. R Moodenbaugh

Abstract—Magnetization measurements show that modern high current-density Nb\(_3\)Sn strands made for HEP programs exhibit flux-jump instabilities at low fields, due to their having large effective filament diameters. Such instabilities might be problematic because they can initiate a quench in low-field regions of magnets. We explored magnetization and transport measurements of the most recent high \( J_c \) Nb\(_3\)Sn strands and cables to probe the instability behavior. In the regime where flux jumps are seen by magnetization measurements, transport current measurements show a threshold for stability. This threshold is significantly lower than the critical current at higher fields, and above this threshold, quenching in the strand could be initiated by ramping the magnetic field. The threshold current depends on the wire size and internal filament design, and is consistent with stability criteria. In cables, quench currents were nearly independent of field after training, and were far below the expected critical currents. Details of these measurements and their implications for testing and use in magnets are discussed.

Index Terms—Electric variables measurements, niobium-tin compounds, stability, superconducting filaments and wires

I. INTRODUCTION

The critical current density, \( J_c \), of Nb\(_3\)Sn wires has increased dramatically over the last several years. Oxford Instruments-Superconducting Technology (OI-ST) has developed internal-Sn design, 0.8 mm diameter wires with a \( J_c \) over the non-copper stabilizer area of 3000 A/mm\(^2\) at 12 T and 4.2 K [1]. Cables produced from these strands have enabled the achievement of 16 T in model accelerator dipole magnets [2]. Such a high \( J_c \) is a result of strand designs that maximize the area of Nb\(_3\)Sn in the “sub-elements”. This usually requires a high content of alloyed Nb filaments and large tin cores and individual sub-element barriers of pure Nb. Also wires are reacted for long periods of time at temperatures of 650-700 °C. During this reaction, the filaments coalesce to a solid mass with significant conversion of the barrier to Nb\(_3\)Sn, thus producing large effective diameters of Nb\(_3\)Sn. This in turn produces large magnetizations at low fields and the occurrence of magnetic instabilities, which is seen as flux-jumps in magnetization measurements [3,4]. An example of this is shown in Fig. 1 for a 2985 A/mm\(^2\), 0.7 mm strand manufactured by OI-ST.

In this paper we describe measurements (first reported at a workshop at Fermilab [5]) which show that these magnetic instabilities can initiate quenches (irreversible transition to the normal state) in these wires carrying transport currents at low fields. A similar instability is also observed in the tests of Rutherford cables.

II. STRAND ELECTRICAL TEST

A. Current-Voltage Sweep

The usual superconductor wire electrical test at 4.2 K consists of measuring the critical current, \( I_c \), by recording the voltage \( V \) across a length of specimen (50-100 cm) as a function of increasing current, resulting in a \( V-I \) curve. \( I_c \) is defined from this curve as the current at which the sample voltage meets an arbitrary criterion, \( e.g. \) when the resistivity is \( 10^{-14} \) Ω·m, as is usual for NbTi wires, or the electric field is 10 \( \mu \)V/m, as is often used for Nb\(_3\)Sn wires. \( I_c \) is determined as a function of field by performing this measurement with a suitable holder placed in a solenoid. The most widely used holder is the ITER barrel arrangement [6]. For “stable” conductors, the limitation of this test is then dictated by the available field and current. The facility at Brookhaven National Laboratory (BNL) has been used for many such tests of stable conductors, up to a maximum field of 11.5 T and a maximum current of 1 kA. The sample heat-treatment,
assembly and test procedures are described in [7].

High $J_c$ Nb$_3$Sn wires are typically "unstable"; i.e. wires tend to quench (irreversible transition to the normal state) before the $I_c$ criterion is reached. In some cases this quench current, $I_{q}$, can be significantly below $I_c$ with no detectable development of voltage. This has been referred to as a lack of "self-field" stability [8]. Often, when faced with this situation, the wires are tested successfully at higher fields where $I_c$ is lower [9]. The stabilizer volume and resistivity, the cooling conditions, and the filamentary nature of the composite have a significant effect on this stability. A point to note is that in most holder designs for Nb$_3$Sn $I_c$ tests, the current transfer from the copper leads to the strand is done in the region of uniform central field of the solenoid.

B. Field-Sweep

An alternate way to approach a quench is to set the current and sweep the field. This was used by Wilson and Walters [10] to study stability of early multifilament composites of NbTi. Sweeping the field is much like what is done in a magnetization measurement, and when a flux-jump occurs, it can create a normal region in the wire which can either recover or grow into a propagating zone and quench the sample. This measurement is more effective in studying "magnetic" instability of a conductor. An example of this is shown in Fig. 2 where the voltage across 110 cm of strand carrying a transport current of 400 A is plotted as the field $H$ is increased from zero in a $V$-$H$ plot. At some field, denoted by $H_{q}$, the sample quenches due to a flux-jump event. Transient data loggers are used to record the quench voltage and current in the sample as shown in Fig. 3. Sometimes, voltage spikes are seen in the $V$-$H$ plot without the sample quenching. In these cases, the normal zone due to a flux-jump recovers to the superconducting state. After a quench the current is re-established and the field ramp is continued typically to ~4T. The sample sometimes quenches at higher fields. A similar trace is taken during the field ramping down to zero. Nominally, the field is ramped at 5 mT/s.

C. Stability Threshold Current Density $J_s$

In this section, we show measurements made on three strands that were manufactured by OI-ST. Samples were all reacted on stainless steel barrels and then transferred to a Ti-alloy barrel for testing in liquid helium. Sample A is a Rod-Restack Process (RRP) wire [1], 0.7mm diameter with a non-Cu fraction of 0.5. The reaction sequence was the following: 48 h @ 210 °C, 48 h @ 340 °C and 180 h @ 675 °C. $J_c$ (12T) for this conductor was 2810 A/mm$^2$. In this case, the power supply limit was reached at 10 T. However, at low fields the stability threshold current density was only 1900 A/mm$^2$. This sample developed more voltage at high fields and also had a higher $J_s$ than Sample A.

Sample B is also a RRP strand at a larger diameter of 0.8 mm. This sample was reacted as follows: 100 h @ 210 °C, 48 h @ 340 °C and 180 h @ 675 °C. $J_c$ (12T) for this conductor was 2810 A/mm$^2$. In this case, the power supply limit was reached at 10 T. However, at low fields the stability threshold current density was only 1900 A/mm$^2$. This sample developed more voltage at high fields and also had a higher $J_s$ than Sample A.

Sample C is a Modified-Jelly-Roll (MJR) strand [11], 0.8 mm diameter, with non-Cu fraction of 0.39. Magnetization measurements of this sample also showed flux-jumps. Its reaction schedule was similar to that for Sample A. The wire had a lower $J_c$ (12 T) of 1825 A/mm$^2$ than the RRP strands, and was also marginally-stable in $V$-$I$ measurements. For this wire, quench currents were measured at lower fields and found to be erratic and lower than the limit of the power supply.

Fig. 3 Example of a quench trace at $H_q$. Quench starts at 0.201 s. Current is shut-off at 0.215 s. The quench starts within the 90cm section of sample which is wound over the TI-alloy barrel. The 110 cm section is at the end of the Ti-alloy barrel section [7].
Previously observed by E. Barzi [9].

than the RRP strands on account of its lower


Similar unstable $V$-$I$ behavior in MJR strands has been previously observed by E. Barzi [9]. $J_s$ for this wire is higher than the RRP strands on account of its lower $J_c$, as shown in Fig. 5.

III. CABLE ELECTRICAL MEASUREMENTS

30-strand Rutherford cables were fabricated at New England Wire from MJR strand from the same billet as Sample C. For this test, two 1.5 m long pieces were reacted in vacuum (100 h @ 210 °C, 48 h @ 340 °C, and 150 h @ 650 °C) in a straight configuration so that no bending strain would be introduced when tested in the short sample holder. Following the reaction, the cable ends were filled with solder over ~15 cm at the return end and ~30 cm at the lead end. The lead-out section of each sample was soldered to NbTi cables which were eventually attached to the current leads of the test holder. The cable samples were vacuum-impregnated with CDT101™ epoxy in a manner similar to that used in magnet coil construction. This bifilar cable composite was assembled in a fixture with a pre-compression of 35 MPa (5 kPSI), and then mounted in the standard short-sample holder of the BNL cable test facility.

As shown in Fig. 5, the cable was trained in the bore of a vertical dipole magnet to a maximum field of 7 T, with the field applied parallel to the wide face of the cable. The field profile is such that the solder joints at the top are in zero field, while the adjacent 20-cm lengths of the cable lie in a gradient from zero to full field. The 60-cm test section below that has uniform field. Voltage taps were soldered to the edge of the cable at the solder joints and near the margins of the test section.

The current polarity is such that the self-field adds to the applied field. The peak self-field is calculated to be 0.09 T/kA. Details of the test facility and procedures are in [12, 13].

A. Current-voltage test

In a manner similar to that for strands, the cable was first energized with a background field of 7 T. The sample trained up from 13.9 kA to 17.7 kA in 25 quenches. The current of 17.7 kA was significantly lower than what was expected from the strand $I_c$ and the number of strands in the cable. All quenches originated between the solder joints, however many quenches were located in the field gradient region above the test section. Subsequent $V$-$I$ measurements at lower background fields showed no significant increase in the quench current. The $V$-$I$ traces also showed voltage spikes, either from conductor motion or due to flux-jumps. The latter speculation is borne out by the incidence and magnitude of the spikes at zero applied field, as shown in Fig. 6.

The cable test is similar to how a magnet would behave. In this test, as in a magnet, there are always regions of conductor at very low fields where flux jumps dominate and can initiate a quench. We confirm this magnetic instability by the tests described below.

B. Field-Sweep Measurements

In analogy with wire tests, $V$-$H$ traces were made at set currents for field sweeps from 0 to 5 T. Not unexpectedly we find similar quenching at low field. An example of this is shown in Fig. 7, where the voltage across the bifilar cable is recorded as a function of field, while ramping at 5 mT/s.

Fig. 8 summarizes the quench behaviour of this cable. From repeated cycles with various set currents, a threshold current is determined, below which the conductor is stable. $J_s$ for the
cable was 2200 A/mm\(^2\). This value is lower than that observed for the strand, which is probably due to the strand being totally exposed to liquid He and has better heat transfer to the coolant, whereas the cable is potted in epoxy and has worse heat transfer. To check this interpretation, another specimen of Sample C was measured, once while exposed to liquid He on the standard test barrel, and again after wrapping the sample and barrel with fiberglass epoxy to exclude He. \(J_s\) dropped from 2855 to 2260 A/mm\(^2\). This indicates that heat transfer to the bath, has some influence on conductor stability.

IV. CONCLUSIONS

From adiabatic stability models, it is evident that the high \(J_c\) and large filament diameter Nb\(_3\)Sn wires would exhibit flux-jump instability. \(V-H\) measurements, which are analogous to magnetization measurements in the presence of transport current, show a threshold current density below which the conductor does not quench. This \(J_s\) is a function of the dynamic stability of the strand which is influenced by heat transfer within the strand and to the coolant bath. The wires and cable described here were all reacted for long times at 650-675 \(^\circ\)C. This results in Sn penetrating through the barrier and reacting with the copper, and low RRR of the copper stabilizer (RRR is the ratio of room temperature resistivity and the residual resistivity at 20 K). RRR for samples A, B and C were 2.9, 10.8, and 4.8 respectively. A low RRR implies a low thermal conductivity and a reduced stability to flux-jumps. In a follow-on study [14], we find that the RRR of the strand has a strong influence on \(J_s\). At an RRR of \(~\)50, the \(J_s\) of sample A increases by more than a factor of two. This is significant, in that by increasing the dynamic stability these modern high \(J_s\) strands can be used reliably in high field magnets without being limited by the lack of magnetic stability.

ACKNOWLEDGMENT

We thank A. Werner and R. Sikora for heat treating the conductors, and J. D’Ambra and P. Philipsberg for their help in testing cables.

REFERENCES